Advanced Materials Research Methods

X Ray Diffraction Part 2

X Ray phase analysis (quantitative) Structural calculations – Vegard`s Rule GID configuration for thin films exploration

Atomic Force Microscopy AFM Principles of operations of microscope Modes of work of AFM Application

The powder X Ray dyfractometry of polycrystalline samples

sample:

 Polycrystalline powder sample of granulation in the range of 0,1 – 10 μm (0,0001 – 0,001 mm),



Solid sample (attention on stress and texture)

radiation:

> monochromatic $K\alpha$ or $K\alpha^1$,

measure equipment:

- > Two axes goniometer
- Bragg-Brentano geometry (most often)

The powder X Ray diffraction pattern of polycrystalline sample



The qualitative phase analysis

inealy (curts)	d 10 15 20 25 30 35 40 45 50 55 2Theta (°)	$h_{kl} = \frac{\lambda}{2\sin\theta}$ $I_{rel} = \frac{I_n}{I_{max}}$	100 x			
Π	14-0696 Wavelength = 1.5405					
	BPO ₄ Boron Phosphate	d (Å) 3.632	Int 100	h 1	k 0	l 1
	Rad.: CuKα1λ: 1.5405Filterd-sp: Guinier 114.6Cut off:Int.: FilmI/lcor.:3.80Ref: De WoIFF. Technisch Physische Dienst.DelftThe Netherlands.ICDD Grant-In-Aid	3.322 3.067 2.254 1.973	4 4 30 2	0 1 1 1	0 1 1 0	2 0 2 3
	Sys.: TetragonalS.G. I $\overline{4}$ (82)a: 4.338b:c: 6.645A:c: β : γ Z: 2mp:	1.862 1.816 1.661 1.534	8 4 1 2	2 2 0 2	1 0 0 2	1 2 4 0
	Ref: Ibid Dx: 2.809 Dm: SS/FOM:F ₁₈ =89(.0102.20)	1.460 1.413 1.393 1.372	- 8 1 1 2	2 3 2 3	- 1 0 2 1	3 1 2 0
	PSC: tl12. To replace 1-519. Deleted by 34-0132. Mwt: 105.78 Volume [CD]: 125.05	1.319 1.271 1.268	4 1 2	2 1 3	0 0 1	4 5 2
		1.211 1.184	2 2	3 3	0 2	3 1
	2002 JCPDS-International Centre for Diffraction Data. All rights reserved. PCPDFWIN v.2.3					

The quantitative phase analysis $J_{hkl} = C \cdot |F_{hkl}|^2 \cdot LP \cdot p \cdot A \cdot V_n \quad \text{in multiphase systems}$

/F_{hkl} / ² - the structure factore containing temperature factor LP - Lorentz and polarity factor (ungle factor) p - the plane multiplicity factor A - the absorbtion

$$\mathbf{C} = \mathbf{J}_{\mathbf{o}} \cdot \lambda^3 \, \mathbf{N}^2 \cdot \left(\frac{\mu_{\mathbf{o}} \, \mathbf{e}^2}{4\pi \, \mathbf{mr}}\right)^2$$

- J_{o} the intensity of primary been
- λ X Ray radiation lenght
- μ_{o} the magnetic permeability in vacuum
- e the electron charge
- m the mass of electron
- r the distance between an electron and the detection point
- $N\,$ the number of unit cells in $\,1\,\,cm^3$
- **V**_n the volume fraction of the n-phase

The absorbtion coeficient

- $J_{\rm o}\,$ the intensity of X Ray beam passing throught the absorbent of dx thickness
- dJ the loss of intensity of X Ray beam passing throught the absorbent, proportional to I_o , dx and μ
- $\boldsymbol{\mu}$ the linear absorbtion coefficient

 $dJ = \mu J_o dx$

Beer equiation of absorbance - μx J = J_o e

 μ = 1/(2A) in a plat samples (in X Ray diffractometers) μ ^{*} - the mass coefficient of absorbtion, μ ^{*} = μ/ρ

X Ray phase analysis - methods

the direct comparison of patterns:

- for two phases of very similar (almost identical) μ^* (a mixture of phases absorbs X Ray in the same way as a single phase);

the internal standard method

-when μ^* of pure, seperate phase and of mixture differ from one another

the external standard method

- when μ^* of pure, seperate phase and of mixture differ from one another

Rietveld refinement method

- the mathematic profile analysis, independent of the diffrences in μ^* values of phases in the mixture (sample)

The internal standard method

 α -Al₂O₃ itp...

$$\begin{split} J_{hkl}{}^{a} &= \underbrace{C \cdot \left| F_{hkl} \right|^{2} \cdot LP \cdot p}_{K_{a}} \cdot A \cdot V_{a} & \text{standard MgO, Si, } \alpha - Al_{2}O_{3} \text{ itp...}}_{K_{a}} \\ A &= 1/(2 \,\mu); \quad \rho_{a} &= m_{a}/V_{a} \longrightarrow V_{A} &= m_{a}/\rho_{a} \\ m_{a} &\longrightarrow X_{a} \quad X_{a} - \% \text{ content of A phase} \\ m_{w} &\longrightarrow X_{w} \quad X_{w} - \% \text{ content of A phase} \\ m_{w} &\longrightarrow X_{w} \quad X_{w} - \% \text{ content of standard} \\ J_{hkl}{}^{a} &= \frac{K_{a}^{\ \cdot \ \cdot \ x_{a}}}{\mu^{*} \cdot \rho_{a}} \quad \text{for A phase} \\ J_{hkl}{}^{a} &= \frac{K_{w}^{\ \cdot \ \cdot \ x_{w}}}{\mu^{*} \cdot \rho_{w}} \quad \text{for standard} \\ J_{hkl}{}^{a} &= \frac{K_{a} \cdot X_{a}}{\mu^{*}} \\ J_{hkl}{}^{a} &= \frac{K_{w} \cdot X_{w}}{\mu^{*} \cdot \rho_{w}} \\ \end{bmatrix}$$

The calculation of phase A content - X_A

We choose one analitycal reflection for:

$$\frac{\mathbf{J}_{hkl}^{a}}{\mathbf{J}_{hkl}^{w}} = \frac{\mathbf{K}_{a} \cdot \mathbf{X}_{a}}{\mathbf{K}_{w} \cdot \mathbf{X}_{w}}$$
$$\frac{\mathbf{J}_{hkl}^{a}}{\mathbf{J}_{hkl}^{w}} = \mathbf{K} \frac{\mathbf{X}_{a}}{\mathbf{X}_{w}}$$

for analyzed phase A J_{hkl}^a (the strongest one)

for standard J_{hkl}^w
 (according to the literature data)

$$X_{a} = \frac{J_{hkl}^{a} X_{w}}{J_{hkl}^{w} K}$$

content of phase A

[%] or the mass fraction

The calibration curve (an example for CaCO₃)

Data		Total amount	mass fractions		Intensities			
CaCO3	AI2O3		CaCO3	AI2O3	CaCO3 int	Al2O3 int	X	у
0,095	0,3923	0,4873	0,194952	0,805048	83,04	71,7	0,242162	1,158159
0,1885	0,3009	0,4894	0,385166	0,614834	128,95	44,36	0,626454	2,906898
0,248	0,2447	0,4927	0,503349	0,496651	150,1	32,98	1,013486	4,551243
0,2862	0,2031	0,4893	0,584917	0,415083	163,6	26,76	1,409158	6,113602
0,4002	0,1113	0,5115	0,782405	0,217595	204,38	11,07	3,595687	18,46251

 $J_a/J_w = f(x_a/x_w)$

linear function y = ax+ba = K (stała K) $b \cong 0$

for CaCO₃: y = 5.2115 x - 0.534 K = 5.2115

The integral intensity – the area under the peak curve



 $\mathbf{J}_{\mathrm{a}}/\mathbf{J}_{\mathrm{w}} = \mathbf{f}(\mathbf{x}_{\mathrm{a}}/\mathbf{x}_{\mathrm{w}})$

The accurancy and errors in the quantitative analysis

Differences between a structure of phases and a standard

Measurement parameters

•the unstable X Ray source

the unstable detector

- the different [F_{hkl}]²
- the different volume of unit t cells
- differences in density
- the possibility of solid solutions presence

Preparation of samples

 the unproper homogenization

the texture of samples

the unproper granulation

 errors in a monochromator operating

•errors in a goniometer operating

Structural calculations – unit cell parameters

Squared equations:

 $1/d_{hkl}^2 = h^2/a^2 + k^2/b^2 + l^2/c^2$ in orthogonal systems

 $1/d_{hkl}^2 = 4/3 [(h^2 + k^2 + hk)/a^2 + l^2/c^2]$ in the hexagonal system



An orthogonal unit cell

 $n\lambda = 2 d_{hkl} sin\theta$

Unit cells in the hexagonal system

Vegard`s Law

Lattice parameters of solid solutions of ionic crystalls vary linearly in relation to the increasing content of the component substituted according to the equatation:

 $a_r = a_1 + (a_2 - a_1) \cdot C_2 / 100$

 a_r – lattice parameter of solid solution a_1 - lattice parameter of a solvent a_2 - lattice parameter of a solute C_2 – content of solute [% mol.]

Diagram: $a_r = f(C_2 / 100)$



Linear nature of diagram confirms the presence of the solid solution and shows its range.

Solid State Solutions





Substitution solid solution

- The similar ion radius (less than 15%)
- The same type of chemical formule
- The same charge
- The same type of lattice
- The similar
 electronegativity

Interstitial solid solution

- Movable ions in interstitial
- positions
- The electroneutrality of crystal

Subtraction solid solution

Solid State Solutions

$$\label{eq:rhor} \begin{split} \rho_r &= X \text{ Ray density (theoretical)} \\ \rho_r &= \frac{A \cdot Z}{V_k} + 1.6602 \cdot 10^{-24} \end{split}$$

A – the molecular weight Z – the number of molecules in the unit cell V_k – the unit cell volume Calculations for the solvent structure $\rho_p
 - the pycnometer density (real)
 (real)$

Substitution solid solution



Interstitial solid solution



Substraction solid solution

 $\rho_r = \rho_p$

 $\rho_r > \rho_p$

Thin films measurements – GID configuration



Thin films deposited on different substrates (steel, titanium alloys, glass etc.) need special measurement parameters in order to reduce the effect from the substrate and strenghten the reflections originating from coating. Then the measurement configuration with the stable incidence angle (ω) is applied.

ω – constant, small angle of incidence of values between 1 and 3 °.

GID

Grazing Incidence Diffraction

Measurements in GID configuration



$\begin{array}{l} \text{Crystallites dimensions- Scherrer`s} \\ \text{equitation} \\ \text{k} \ \lambda \\ \\ \text{hkl} &= \frac{\beta \ \cos \theta}{} \end{array}$

where:

- β the full width at half maximum $\beta = \beta_{obs} \beta_{stand}$, [rad]
- λ the lengh of radiation applied, $\lambda = 1.5406$ [Å]
- k Scherrer constant, in the range of 0.9 1.0, k = 0.9
- D_{hkl} the average crystallite size (the dimension perpendicular to the plane which gave the reflection)

Where to apply different peak parameters ?



Scanning Probe Microscopy SPM

Scanning Tunneling Microscopy STM

Atomoc Force Microscopy AFM

Application of STM

1. Imaging of the sample surface is based on the applying of the tunneling current between the tip (the probe is a metal needle) and the sample. So the sample has to be conductive or covered by the conduktive coating.

2. The highest resolution mode of SPM (atomic resolution level)

Applications of AFM

- 1. Imaging of surface of the sample the only limit is very !!! clean surface
- **2.Exploration of some properties distribution:**
- frictional forces
- an adhesion
- the spatial distribution of magnetization
- the spatial distribution of electric charge (the surface potiental)
- 3. The modyfication of local properties of sample
- nanolithography



Scheme of AFM Microscope type: Multimode 8.0 by Bruker

AFM operation is based on the interactions between the probe and the surface of sample which cause the deformation of probe, detected by the optical systems.

Exploration of surface in tAFM - how it works?

Image is obtained by the detection of forces interacting on the probe



Forces causing the deformation of the probe are detected by the optical laser system (photodetector)

Scanner - piezoelectric tube with six electrodes which enable the tube to move in three directions XYZ. The scanner allows to move the sample under the tip (usually) in XY directions, in proper distance corelating with the strenght of impact and with the deformation of the probe. The motion of the scanner in Z direction, closely connected with the probe deformation, reflects the changes in the profile of the surphace.

Sondy



Probe - tip, cantilever, needle, etc.

Parameters:

- lenght from 100 to 500 μm,
- curve radius about 2 nm

(also 1 nm in high resolution measurements or even 20 nm in case of measurements in liquid cells)

- Force constants 0.01 1 N/m
- Resonant frequencies in the range of 2 120 kHz



Different types of probes



Source: Bruker 2013

AFM Microscopy

The type of interactions between the tip and the surface atoms influenses on the surface topography and the surface properties research:

- > the local potential distribution research
- the local magnetic field distribution research
- > Young Modulus distribution research
- the topography research:
 - Van der Walls forces
 - The deformation of the probe according to Hook law
 - Interactions with the water layer on the sample surface

Modes of AFM operations

Modes of operation:

✓ Contact;

- ✓ LiftMode (no contact)
- ✓ Semi-contact:

Tapping

Peak Force Tapping (Scan Assist)

Contact mode



- ✓ First mode applied in AFM
- ✓ Through all measurement, the strenght of impact is constant
- Every change of this force cases the scanner to move up or down in Z direction to come back to primary value of strenght.
- ✓ Mode applied to hard and very hard materials

Semi-contact modes: -Tapping -Peak Force Tapping



- ✓ The probe vibrates with the constant frequency and amplitude
- ✓ When any change on the surface appears, the amplitude of probe vibrations also changes.
- ✓ Then the system send a signal to the scanner to move itself in Z direction (up or dawn) until the probe will vibrate with the promary amlitude and frequency.
- ✓ The recorded motion of scanner in Z directions reflects changes in surface profile.



✓ Constant distance between the tip and surface what ce be treated as the "zero" strength of impact

 $\checkmark Z$ direction motion of scanner correlates the changes in the shape of surface

✓ Van der Walls forces are main to influance on the probe deformation.

TiO₂ –Al₂O₃ thin film synthesized by sol-gel method and deposited on steel by dip coating technique.



TiO₂ –Al₂O₃ thin film deposited on steel and incubated in SBF for 30 days





Carbon tubes



Coatings containing SiO₂ (silsesquisilicates) deposited in epitaxial beam

2.6 nm

-1.8 nm



200.0 nm





TiO₂ –Al₂O₃ /Ag thin film synthesized by sol-gel method and deposited on steel by dip coating technique.









0.0

Height

1.0 um

SiO₂ (14 % weight) thin film synthesized by sol-gel method and deposited on steel by dip coating technique -section





-2.1 nm





SiO₂ (10 % weight) thin film synthesized by sol-gel method and deposited on steel by dip coating technique -section





